

## The Crystal Structure of Some Hexakis(methyl cyanide)metal(II) Tetrachlorometallates(III), $M(II)(NCCH_3)_6[M(III)Cl_4]_2$

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Crystals of  $Fe^{II}(NCCH_3)_6[Fe^{III}Cl_4]_2$  are trigonal with space group  $P3$ . The lattice dimensions are  $a = 11.682$  (6) and  $c = 6.134$  (4) Å. Diffractometer data (Mo  $K\alpha$  radiation) were used. After correction for twinning (twinning percentage 14.1) anisotropic refinement with 1650 observed reflexions resulted in an  $R_w$  value of 0.047. The structure is composed of rows of  $(FeCl_4)^-$  tetrahedra and  $Fe(NCCH_3)_6^{2+}$  octahedra. A number of interatomic distances are given. The isomorphous complex  $Mg(NCCH_3)_6(AlCl_4)_2$  (photographic Weissenberg data) is discussed and the related compound  $Ni(NCCH_3)_6(GaCl_4)_2$  (precession data) is briefly mentioned.

### Introduction

Complexes  $M(II)(NCCH_3)_6[M(III)Cl_4]_2$  [hereafter  $M(II)-M(III)$ ], in which  $M(II)$  is Mg, Cd or a divalent metal of the first transition series and  $M(III)$  is B, Al, Fe, Ga, In or Tl, were investigated by Reedijk & Groeneveld (1968*a*). X-ray powder patterns show that, with the exception of Cu compounds (forming a class by themselves), three different types of structure (*A*, *B*, *C*) occur in this series. Mössbauer measurements, infrared spectra (Reedijk & Groeneveld, 1968*b*), magnetic susceptibilities and e.p.r. spectra (Reedijk, 1969) indicate that the  $M(II)(NCCH_3)_6^{2+}$  and  $M(III)Cl_4^-$  ions have octahedral and tetrahedral shapes respectively, but an exact assignment of the molecular symmetry could not be achieved with these techniques.

It was thought interesting to investigate the exact geometry and symmetry of these complex ions by means of X-ray diffraction, and to establish the relationship between the different polymorphs and the sizes and symmetries of the anions and cations. Since most of the complexes are highly sensitive to moisture, the choice of compounds suitable for X-ray work is limited. The analysis was started with  $Mg(NCCH_3)_6(AlCl_4)_2$  (type *A*), this complex containing relatively light metal atoms. Although its structure could be refined to a reasonable degree of accuracy ( $R = 0.18$ ), we encountered a number of inconsistencies between calculated and observed structure factors, perhaps due to disorder phenomena (see below). Therefore we decided to analyse  $Fe(NCCH_3)_6(FeCl_4)_2$ , also belonging to type *A*.

As an example of a *C*-type structure we investigated crystals of  $Ni(NCCH_3)_6(GaCl_4)_2$ . Its structure is related to that of *A*, but so far no solution could be found. Closely related to type *A* is the structure of  $Fe(NCH)_6(FeCl_4)_2$  proposed by Constant, Daran & Jeannin (1970). Brief comments on the two structures in relation to the *A*-type structure are given in the present

article. The *B*-type structure displays a very intricate powder diffraction pattern and has, so far, not been investigated by us.

### Experimental

The complexes were obtained by dissolving stoichiometric amounts of the divalent and trivalent metal chlorides in an excess of methyl cyanide. The Mg-Al compound was prepared using the adduct  $AlCl_3 \cdot 2CH_3CN$  instead of  $AlCl_3$  (Reedijk, 1968). After cooling, the complexes crystallized. The type *A* Mg-Al and Fe-Fe compounds consist of white and green-yellow, hexagonally prismatic crystals respectively. The type *C* Ni-Ga compound consists of pale blue thin hexagonal plates. The crystals were filtered, washed with cold methyl cyanide and dried *in vacuo*. Because of their extreme sensitivity to moisture the crystals were inserted into glass capillaries. After immersion in liquid paraffin, which was dried several times with  $P_2O_5$ , the capillaries were sealed.

Cell dimensions of the Mg-Al compound were determined from zero-layer Weissenberg photographs taken about [010] and [001] with unfiltered copper radiation ( $\lambda_{K\alpha 1} = 1.54051$  Å,  $\lambda_{K\alpha 2} = 1.54433$  Å,  $\lambda_{K\beta} = 1.39217$  Å) superposed with aluminum powder lines ( $a = 4.0492$  Å at 20°C) for calibration purposes. Non-integrated intensity data were obtained with the equi-inclination Weissenberg method (multiple-film technique) from crystals mounted about [001] and [010] using Ni-filtered Cu  $K\alpha$  radiation. Intensities from six levels about [001] ( $l = 0$  up to 5) and ten levels about [010] ( $k = 0$  up to 9) were estimated visually and reduced to structure-factor moduli in the usual way. An absorption correction was applied to the *c* axis data, assuming a cylindrical cross-section for the crystal used ( $r \approx 0.02$  cm,  $\mu_r = 1.6$ ). On the basis of the *b*-axis data, the *c*-axis data comprising 821 symmetry-independent reflexions (including 126 non-observed ones) were put on a com-

mon scale. Non-observed reflexions were given intensity values equal to the lowest measured intensity.

The lattice parameters, as well as the X-ray intensity data of the Fe-Fe compound, were measured on a Nonius three-circle diffractometer using Zr-filtered Mo  $K\alpha$  radiation ( $\lambda=0.71069$  Å). According to the classification rules of Arndt & Willis (1966) the goniostat has normal-beam equatorial diffraction geometry, with  $\chi$ -motion restricted between 0 and  $-90^\circ$ . A hexagonally prismatic crystal with approximate dimensions  $0.15 \times 0.15 \times 0.6$  mm was mounted about [001]. The intensities were collected by means of the  $\theta$ - $2\theta$  scan mode to a maximum  $\theta$  value of  $35^\circ$ ; high intensities were reduced by using attenuation filters. The background intensity was measured at each side of a reflexion for half the scanning time. A reflexion was considered significant if the net count exceeded twice the standard deviation. The number of observed symmetry-dependent reflexions having positive glancing angles was 2374 and that with negative values 341. An absorption correction was applied according to the Monte Carlo method as programmed by de Graaff (1972). The repeated measurements of a number of standard reflexions ensured the correction for the decrease in scattering power of the crystal by means of a polynomial function of the exposure time. An average was taken of the symmetry-dependent reflexions

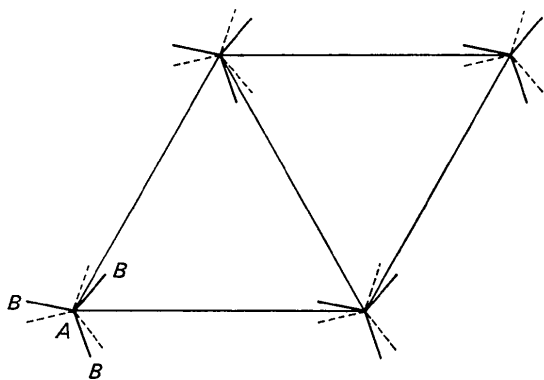


Fig. 1. A simplified model for disorder or twinning: the orientation of  $AB_3$  groups about the threefold rotation axis  $z$  in the Fe-Fe structure. The solid lines alone represent the ordered structure; the solid and dashed lines in statistical distribution of the disorder; the solid lines in one part of the lattice and the dashed lines in another part of the crystal describe the twinned structure.

and all data were reduced to structure factors in the usual way. A total of 1755 observed reflexions remained including 266 pairs  $hkl$  and  $\bar{h}\bar{k}\bar{l}$ , and 595 non-observed ones including 31 pairs  $hkl$  and  $\bar{h}\bar{k}\bar{l}$ .

The cell dimensions of the Ni-Ga compound were determined with Mo  $K\alpha$  radiation from  $hk0$  and  $0kl$  precession photographs. Film shrinkage was corrected for.

Both the symmetry relations in reciprocal space and the Patterson function of the compounds Mg-Al and Fe-Fe exclude all space groups except  $P3$  and  $P\bar{3}$ . For the present the space group  $P3$  was assumed; an  $R_w$ -ratio test (Hamilton, 1965) justified this assumption. Crystals of the Mg-Al compound floated on  $CCl_4$  (density of  $1.6$  g.cm $^{-3}$ ). Hence the unit cell contains one formula unit Mg-Al.

The Ni-Ga complex has the space group  $P3$  or  $P\bar{3}$ . The cell volume is about six times that of the Fe-Fe compound [ $a(\text{Ni-Ga}) \simeq a(\text{Fe-Fe})/3$ ;  $c(\text{Ni-Ga}) \simeq 2c(\text{Fe-Fe})$ ]. Reflexions with  $h-k=3n \pm 1$  are not present if  $l=0$  or  $l=2n+1$ , and are always weak; this indicates a subcell with dimensions of the Fe-Fe compound. Crystal data for the three compounds are given in Table 1.

#### Structure determination and refinement

The presence of one formula unit of the M(II)-M(III) compound in a unit cell with symmetry  $P3$  requires that the three metal ions and two chlorine ions Cl(2) and Cl(4) be in the special positions  $0,0,z$ ,  $\frac{1}{3},\frac{2}{3},z$  and  $\frac{2}{3},\frac{1}{3},z$ . The remaining ions Cl(1), Cl(3) and the ligand atoms are accommodated in general positions. An approximate model for the structure of the Mg-Al compound was derived from a Patterson map and successive Fourier syntheses. The Patterson function of the Fe-Fe compound showed that the two structures were isomorphous.

During the least-squares refinement of the Fe-Fe and Mg-Al structures, the following scattering factors were used: (1) the values for  $Fe^{2+}$ ,  $Fe^{3+}$  and  $Cl^-$  ions evaluated by Cromer & Waber (1965), (2) those for  $Mg^{2+}$ ,  $Al^{3+}$ , N and C (valence) from *International Tables for X-ray Crystallography* (1962), all values except those for nitrogen and carbon being corrected for the real part of the anomalous scattering. The imaginary part of the scattering factor was not taken into account.

Table 1. Unit-cell parameters of three compounds  $M(\text{II})(\text{NCCH}_3)_6[\text{M}(\text{III})\text{Cl}_4]_2$

The numbers in brackets are estimated standard deviations in terms of the last decimal given.

M(II)-M(III)	Mg-Al	Fe-Fe	Ni-Ga
Space group	$P3$	$P3$	$P3$ or $P\bar{3}$
$a$	11.693 (5)	11.682 (6)	20.526 (4) Å
$c$	6.151 (3)	6.134 (4)	11.879 (3) Å
$d(\text{calc})$	1.387	1.597	1.637 g.cm $^{-3}$
$V$	728.3	725.0	4334.4 Å $^3$
$Z$	1	1	6
$\mu$	86 (Cu $K\alpha$ )	22.5 (Mo $K\alpha$ )	33.4 cm $^{-1}$ (Mo $K\alpha$ )



disorder the distribution is statistical and would result in the symmetry  $P31m$  for a 1:1 arrangement. The twinned structure arises from the configuration  $x, y, z$  in one part of the lattice and the configuration  $y, x, z$  in another part. In this case (100) acts as mirror plane for the separate individuals. The justification for these assumptions can be seen in Fig. 2, the ultimate structure. All atoms in general positions are located near one of the planes  $\{30\bar{3}0\}$  and only small rotations are required to obtain the mirror configuration. In both models an extra parameter, the occupation number  $p$  ( $0 \leq p \leq 1$ ), is introduced for occupation of position  $x, y, z$  with contribution  $1-p$  and position  $y, x, z$  with contribution  $p$ .

### (a) Disorder

The contributions to the calculated structure factor  $F_c'$  are coherent and have to be added according to

$$\begin{aligned} F_c' &= F_c(x, y, z) + F_c(y, x, z) \\ &= (1-p) \sum_{j=1}^N f_j \exp [2\pi i(hx_j + ky_j + lz_j)] \\ &\quad + p \sum_{j=1}^N f_j \exp [2\pi i(hy_j + kx_j + lz_j)]. \end{aligned} \quad (1)$$

Corresponding to the choice of origin  $(0,0,0, \frac{2}{3}, \frac{1}{3}, 0)$ , or  $(\frac{1}{3}, \frac{2}{3}, 0)$  there are three possibilities for introducing disorder by means of a pseudo-mirror plane: through the  $\text{Fe}(\text{NCCH}_3)_6$  octahedron, or through either one of the two  $\text{FeCl}_4$  tetrahedra. Difference maps calculated after isotropic and anisotropic 'ordered' refinements indicated that the first possibility (exchange of the two  $\text{FeCl}_4$  tetrahedra) was unlikely.

The refinement was executed with coupled atoms at  $x, y, z$  and  $y, x, z$ , having equal isotropic thermal par-

Table 2 (cont.)

H	K	L	F0	F1	F2	F3	F4	F5	F6	F7	F8	F9	F10	F11	F12	F13	F14	F15	F16	F17	F18	F19	F20	F21	F22	F23	F24	F25	F26	F27	F28	F29	F30	F31	F32	F33	F34	F35	F36	F37	F38	F39	F40	F41	F42	F43	F44	F45	F46	F47	F48	F49	F50	F51	F52	F53	F54	F55	F56	F57	F58	F59	F60	F61	F62	F63	F64	F65	F66	F67	F68	F69	F70	F71	F72	F73	F74	F75	F76	F77	F78	F79	F80	F81	F82	F83	F84	F85	F86	F87	F88	F89	F90	F91	F92	F93	F94	F95	F96	F97	F98	F99	F100	F101	F102	F103	F104	F105	F106	F107	F108	F109	F110	F111	F112	F113	F114	F115	F116	F117	F118	F119	F120	F121	F122	F123	F124	F125	F126	F127	F128	F129	F130	F131	F132	F133	F134	F135	F136	F137	F138	F139	F140	F141	F142	F143	F144	F145	F146	F147	F148	F149	F150	F151	F152	F153	F154	F155	F156	F157	F158	F159	F160	F161	F162	F163	F164	F165	F166	F167	F168	F169	F170	F171	F172	F173	F174	F175	F176	F177	F178	F179	F180	F181	F182	F183	F184	F185	F186	F187	F188	F189	F190	F191	F192	F193	F194	F195	F196	F197	F198	F199	F200	F201	F202	F203	F204	F205	F206	F207	F208	F209	F210	F211	F212	F213	F214	F215	F216	F217	F218	F219	F220	F221	F222	F223	F224	F225	F226	F227	F228	F229	F230	F231	F232	F233	F234	F235	F236	F237	F238	F239	F240	F241	F242	F243	F244	F245	F246	F247	F248	F249	F250	F251	F252	F253	F254	F255	F256	F257	F258	F259	F260	F261	F262	F263	F264	F265	F266	F267	F268	F269	F270	F271	F272	F273	F274	F275	F276	F277	F278	F279	F280	F281	F282	F283	F284	F285	F286	F287	F288	F289	F290	F291	F292	F293	F294	F295	F296	F297	F298	F299	F300	F301	F302	F303	F304	F305	F306	F307	F308	F309	F310	F311	F312	F313	F314	F315	F316	F317	F318	F319	F320	F321	F322	F323	F324	F325	F326	F327	F328	F329	F330	F331	F332	F333	F334	F335	F336	F337	F338	F339	F340	F341	F342	F343	F344	F345	F346	F347	F348	F349	F350	F351	F352	F353	F354	F355	F356	F357	F358	F359	F360	F361	F362	F363	F364	F365	F366	F367	F368	F369	F370	F371	F372	F373	F374	F375	F376	F377	F378	F379	F380	F381	F382	F383	F384	F385	F386	F387	F388	F389	F390	F391	F392	F393	F394	F395	F396	F397	F398	F399	F400	F401	F402	F403	F404	F405	F406	F407	F408	F409	F410	F411	F412	F413	F414	F415	F416	F417	F418	F419	F420	F421	F422	F423	F424	F425	F426	F427	F428	F429	F430	F431	F432	F433	F434	F435	F436	F437	F438	F439	F440	F441	F442	F443	F444	F445	F446	F447	F448	F449	F450	F451	F452	F453	F454	F455	F456	F457	F458	F459	F460	F461	F462	F463	F464	F465	F466	F467	F468	F469	F470	F471	F472	F473	F474	F475	F476	F477	F478	F479	F480	F481	F482	F483	F484	F485	F486	F487	F488	F489	F490	F491	F492	F493	F494	F495	F496	F497	F498	F499	F500	F501	F502	F503	F504	F505	F506	F507	F508	F509	F510	F511	F512	F513	F514	F515	F516	F517	F518	F519	F520	F521	F522	F523	F524	F525	F526	F527	F528	F529	F530	F531	F532	F533	F534	F535	F536	F537	F538	F539	F540	F541	F542	F543	F544	F545	F546	F547	F548	F549	F550	F551	F552	F553	F554	F555	F556	F557	F558	F559	F560	F561	F562	F563	F564	F565	F566	F567	F568	F569	F570	F571	F572	F573	F574	F575	F576	F577	F578	F579	F580	F581	F582	F583	F584	F585	F586	F587	F588	F589	F590	F591	F592	F593	F594	F595	F596	F597	F598	F599	F600	F601	F602	F603	F604	F605	F606	F607	F608	F609	F610	F611	F612	F613	F614	F615	F616	F617	F618	F619	F620	F621	F622	F623	F624	F625	F626	F627	F628	F629	F630	F631	F632	F633	F634	F635	F636	F637	F638	F639	F640	F641	F642	F643	F644	F645	F646	F647	F648	F649	F650	F651	F652	F653	F654	F655	F656	F657	F658	F659	F660	F661	F662	F663	F664	F665	F666	F667	F668	F669	F670	F671	F672	F673	F674	F675	F676	F677	F678	F679	F680	F681	F682	F683	F684	F685	F686	F687	F688	F689	F690	F691	F692	F693	F694	F695	F696	F697	F698	F699	F700	F701	F702	F703	F704	F705	F706	F707	F708	F709	F710	F711	F712	F713	F714	F715	F716	F717	F718	F719	F720	F721	F722	F723	F724	F725	F726	F727	F728	F729	F730	F731	F732	F733	F734	F735	F736	F737	F738	F739	F740	F741	F742	F743	F744	F745	F746	F747	F748	F749	F750	F751	F752	F753	F754	F755	F756	F757	F758	F759	F760	F761	F762	F763	F764	F765	F766	F767	F768	F769	F770	F771	F772	F773	F774	F775	F776	F777	F778	F779	F780	F781	F782	F783	F784	F785	F786	F787	F788	F789	F790	F791	F792	F793	F794	F795	F796	F797	F798	F799	F800	F801	F802	F803	F804	F805	F806	F807	F808	F809	F810	F811	F812	F813	F814	F815	F816	F817	F818	F819	F820	F821	F822	F823	F824	F825	F826	F827	F828	F829	F830	F831	F832	F833	F834	F835	F836	F837	F838	F839	F840	F841	F842	F843	F844	F845	F846	F847	F848	F849	F850	F851	F852	F853	F854	F855	F856	F857	F858	F859	F860	F861	F862	F863	F864	F865	F866	F867	F868	F869	F870	F871	F872	F873	F874	F875	F876	F877	F878	F879	F880	F881	F882	F883	F884	F885	F886	F887	F888	F889	F890	F891	F892	F893	F894	F895	F896	F897	F898	F899	F900	F901	F902	F903	F904	F905	F906	F907	F908	F909	F910	F911	F912	F913	F914	F915	F916	F917	F918	F919	F920	F921	F922	F923	F924	F925	F926	F927	F928	F929	F930	F931	F932	F933	F934	F935	F936	F937	F938	F939	F940	F941	F942	F943	F944	F945	F946	F947	F948	F949	F950	F951	F952	F953	F954	F955	F956	F957	F958	F959	F960	F961	F962	F963	F964	F965	F966	F967	F968	F969	F970	F971	F972	F973	F974	F975	F976	F977	F978	F979	F980	F981	F982	F983	F984	F985	F986	F987	F988	F989	F990	F991	F992	F993	F994	F995	F996	F997	F998	F999	F1000	F1001	F1002	F1003	F1004	F1005	F1006	F1007	F1008	F1009	F1010	F1011	F1012	F1013	F1014	F1015	F1016	F1017	F1018	F1019	F1020	F1021	F1022	F1023	F1024	F1025	F1026	F1027	F1028	F1029	F1030	F1031	F1032	F1033	F1034	F1035	F1036	F1037	F1038	F1039	F1040	F1041	F1042	F1043	F1044	F1045	F1046	F1047	F1048	F1049	F1050	F1051	F1052	F1053	F1054	F1055	F1056	F1057	F1058	F1059	F1060	F1061	F1062	F1063	F1064	F1065	F1066	F1067	F1068	F1069	F1070	F1071	F1072	F1073	F1074	F1075	F1076	F1077	F1078	F1079	F1080	F1081	F1082	F1083	F1084	F1085	F1086	F1087	F1088	F1089	F1090	F1091	F1092	F1093	F1094	F1095	F1096	F1097	F1098	F1099	F1100	F1101	F1102	F1103	F1104	F1105	F1106	F1107	F1108	F1109	F1110	F1111	F1112	F1113	F1114	F1115	F1116	F1117	F1118	F1119	F1120	F1121	F1122	F1123	F1124	F1125	F1126	F1127	F1128	F1129	F1130	F1131	F1132	F1133	F1134	F1135	F1136	F1137	F1138	F1139	F1140	F1141	F1142	F1143	F1144	F1145	F1146	F1147	F1148	F1149	F1150	F1151	F1152	F1153	F1154	F1155	F1156	F1157	F1158	F1159	F1160	F1161	F1162	F1163	F1164	F1165	F1166	F1167	F1168	F1169	F1170	F1171	F1172	F1173	F1174	F1175	F1176	F1177	F1178	F1179	F1180	F1181	F1182	F1183	F1184	F1185	F1186	F1187	F1188	F1189	F1190	F1191	F1192	F1193	F1194	F1195	F1196	F1197	F1198	F1199	F1200	F1201	F1202	F1203	F1204	F1205	F1206	F1207	F1208	F1209	F1210	F1211	F1212	F1213	F1214	F1215	F1216	F1217	F1218	F1219	F1220	F1221	F1222	F1223	F1224	F1225	F1226	F1227	F1228	F1229	F1230	F1231	F1232	F1233	F1234	F1235	F1236	F1237	F1238	F1239	F1240	F1241	F1242	F1243	F1244	F1245	F1246	F1247	F1248	F1249	F1250	F1251	F1252	F1253	F1254	F1255	F1256	F1257	F1258	F1259	F1260	F1261	F1262	F1263	F1264	F1265	F1266	F1267	F1268	F1269	F1270	F1271	F1272	F1273	F1274	F1275	F1276	F1277	F1278	F1279	F1280	F1281	F1282	F1283	F1284	F1285	F1286	F1287	F1288	F1289	F1290	F1291	F1292	F1293	F1294	F1295	F1296	F1297	F1298	F1299	F1300	F1301	F1302	F1303	F1304	F1305	F1306	F1307	F1308	F1309	F1310	F1311	F1312	F1313	F1314	F1315	F1316	F1317	F1318	F1319	F1320	F1321	F1322	F1323	F1324	F1325	F1326	F1327	F1328	F1329	F1330	F1331	F1332	F1333	F1334	F1335	F1336	F1337	F1338	F1339	F1340	F1341	F1342	F1343	F1344	F1345	F1346	F1347	F1348	F1349	F1350	F1351	F1352	F1353	F1354	F1355	F1356	F1357	F1358	F1359	F1360	F1361	F1362	F1363	F1364	F1365	F1366	F1367	F1368	F1369	F1370	F1371	F1372	F1373	F1374	F1375	F1376	F1377	F1378	F1379	F1380	F1381	F1382	F1383	F1384	F1385	F1386	F1387	F1388	F1389	F1390</
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The twinning parameter is calculated from

$$p = \frac{1}{2} - \frac{1}{2} \frac{F_o'^2(hkl) - F_o'^2(khl)}{F_c'^2(hkl) - F_c'^2(khl)} \quad (4)$$

It must be pointed out that equations (3) and (4) are only good approximations if  $p$  is small, since the initially calculated structure factors based upon an ordered model are biased by the twinning. From the foregoing treatment of assumed disorder  $p$  is expected to be smaller than, say, 0.20.

Using (4) we are now able to determine  $F_c'$  for the twinned structure:

$$F_c'(hkl) = [(1-p)F_c^2(hkl) + pF_c^2(khl)]^{1/2} \quad (5)$$

Assuming that there are no errors whatsoever in  $F_o$  and  $F_c$ , the following expression applies

$$F_o - F_o' = F_c - F_c' \quad (6)$$

and, accordingly, the observed structure factor can be corrected for twinning by adding the difference  $F_c - F_c'$  to its observed value. With the new set of  $F_o$  values, the refinement of the structure was continued to obtain a better model. Repeating the procedure,  $p$  was redetermined using (4) and better values of  $F_c'$  and  $F_o$  were obtained using (5) and (6). The twinning parameter converged after five cycles to  $p = 0.156$ .

According to a procedure explained in the Appendix all symmetry-averaged reflexions were corrected for twinning and given new weights. It was now possible to resume the least-squares refinement. Compar-

ison of  $F_o$  and  $F_c$  after three cycles with isotropic temperature factors gave an indication that the twinning ratio was too high. After a few trials we accepted  $p = 0.141$  as the best value.

The refinement was continued with anisotropic vibrational parameters. It indicated that the strongest 16 reflexions had too high weights. These reflexions were assigned standard errors three times their original values. The refinement converged towards  $R = 0.049$  (omitting non-observed reflexions) and  $R_w = 0.047$ .

With two exceptions, shifts were smaller than 0.4 times the standard errors. The shifts for C(3) and C(4) equalled the standard deviations.

Hitherto the choice of space group was not settled. We inspected the centrosymmetric structure with Fe(II) at the origin and average positions for the other atoms. Strictly speaking, we should have redetermined the parameter  $p$ , but we accepted the value 0.141, since the centric structure differs slightly from the acentric one and, moreover,  $p$  merely showed minor variations during the trials with the various disorder and twinning models. The anisotropic refinement converged for  $P\bar{3}$  towards  $R = 0.054$  and  $R_w = 0.066$ . It can be seen (Table 6) that the centrosymmetric structure can be rejected on a 0.005 significance level.

As a result of twinning, part of the structure passes into its enantiomorphic form. A determination of the absolute configuration was, therefore, not carried out.

The observed structure factors corrected for a twinning factor of 0.141 and their corresponding calculated

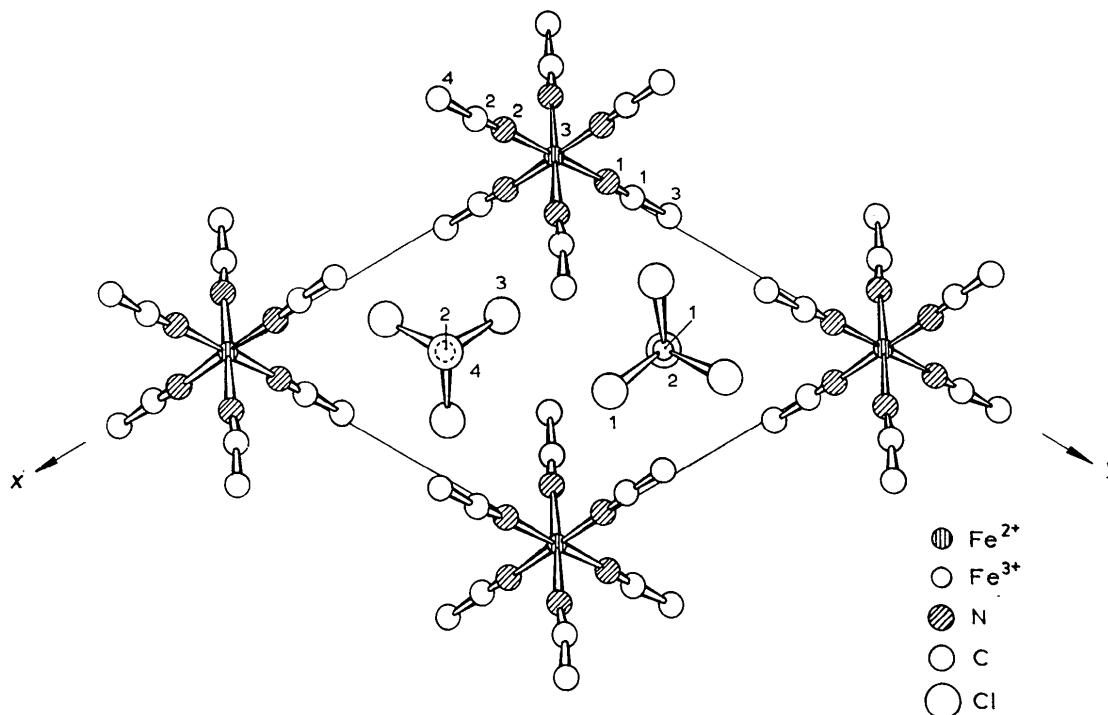


Fig. 2. Projection on (001) plane.

structure factors are listed in Table 2. The positional parameters and their standard deviations are given in Table 3 and the vibrational parameters,  $U_{ij}$ , in Table 4.

Table 3. Atomic parameters (in fractions of cell edges) and their standard deviations (Å)

	$x$	$\sigma(x)$	$y$	$\sigma(y)$	$z$	$\sigma(z)$
Fe(1)	$\frac{1}{2}$	0.0	$\frac{2}{3}$	0.0	-0.0420	0.006
Fe(2)	$\frac{2}{3}$	0.0	$\frac{1}{3}$	0.0	0.0442	0.006
Fe(3)	0.0	0.0	0.0	0.0	0.0	0.0
Cl(1)	0.5201	0.004	0.6888	0.003	0.0631	0.006
Cl(2)	$\frac{1}{2}$	0.0	$\frac{2}{3}$	0.0	-0.4011	0.006
Cl(3)	0.4872	0.005	0.3290	0.004	-0.0740	0.006
Cl(4)	$\frac{2}{3}$	0.0	$\frac{1}{3}$	0.0	0.3997	0.007
N(1)	-0.0124	0.010	0.1460	0.010	-0.1843	0.010
N(2)	0.0081	0.012	-0.1452	0.012	0.2289	0.012
C(1)	-0.0103	0.012	0.2258	0.013	-0.2914	0.014
C(2)	0.0219	0.012	-0.2192	0.011	0.3069	0.010
C(3)	-0.0183	0.016	0.3307	0.014	-0.4320	0.015
C(4)	0.0271	0.016	-0.3266	0.013	0.4229	0.012

Table 4. Vibrational parameters  $U_{ij}$  (Å<sup>2</sup>) ( $\times 10^3$ ) in the temperature factor  $\exp[-2\pi^2 \sum U_{ij} a_i^* a_j^* h_i h_j]$

The estimated standard deviations in digits of the last figure are given in parentheses.

	$U_{11}$	$U_{22}$	$U_{33}$	$2U_{12}$	$2U_{23}$	$2U_{13}$
Fe(1)	47 (1)	47 (1)	39 (2)	47 (1)	0	0
Fe(2)	45 (1)	45 (1)	40 (2)	45 (1)	0	0
Fe(3)	37 (1)	37 (1)	49 (1)	37 (1)	0	0
Cl(1)	43 (1)	91 (2)	73 (2)	71 (3)	-45 (3)	-31 (3)
Cl(2)	86 (3)	86 (3)	30 (3)	86 (3)	0	0
Cl(3)	70 (2)	94 (2)	54 (2)	103 (4)	10 (3)	-11 (3)
Cl(4)	76 (3)	76 (3)	39 (3)	76 (3)	0	0
N(1)	46 (5)	31 (4)	36 (3)	32 (7)	15 (6)	-8 (6)
N(2)	63 (6)	52 (5)	58 (6)	73 (9)	5 (8)	37 (8)
C(1)	25 (4)	54 (6)	80 (8)	10 (8)	-80 (12)	1 (9)
C(2)	41 (5)	46 (5)	40 (5)	52 (8)	46 (7)	9 (7)
C(3)	76 (9)	56 (8)	94 (10)	100 (14)	9 (13)	-12 (14)
C(4)	63 (8)	64 (8)	55 (6)	42 (12)	87 (12)	23 (12)

In the space group  $P3$  one atom must have a fixed  $z$  parameter. As a result this atom, Fe(3), will have zero standard deviation in its  $z$  parameter if full-matrix refinement is applied.

## II. The Mg-Al compound

In the anisotropic refinement of the ordered structure (space group  $P3$ ) disappointingly high indices  $R=0.18$  and  $R_w=0.21$  were attained. Several models for disorder were examined. The best model is a disordered arrangement with Cl(4) at 0,0,0 and a partial exchange of Mg(NCCH<sub>3</sub>)<sub>6</sub> and AlCl<sub>4</sub> groups. The iteration with anisotropic vibrational parameters for the chlorine atoms resulted in  $R=0.150$ ,  $R_w=0.166$  and  $p=0.322$ . The stumbling block, 012, still showed bad agreement and a number of large  $B$  values in the range between 7 and 11 Å<sup>2</sup> were observed for a number of atoms. Nevertheless, the overall agreement had increased substantially. Accepting this model as zero

hypothesis, the ordered structure is rejected on a 0.005 significance level.

Since no improvement was obtained with the twinned model as investigated for the Fe-Fe compound, we accepted the disordered model with Cl(4) at 0,0,0 and  $p=0.322$ . Its geometrical entities are listed in Table 5.

Table 5. Distances (Å) and angles (°) in Me(III)Cl<sub>4</sub> tetrahedra and Me(II) (NCCH<sub>3</sub>)<sub>6</sub> octahedra

Estimated standard deviations in digits of the last decimal are given in parentheses.  $a$ : Fe-Fe compound with twinning parameter  $p=0.141$ ;  $b$ : the same as  $a$  but with constraint of equal Fe(3)-N distances;  $c$ : disordered Mg-Al compound with occupation number  $p=0.322$ .

Fe(1)/Al(1) tetrahedron	$a$	$b$	$c$
Fe(1)-Cl(1)	2.163 (4)	2.164 (4)	2.15 (2)
Fe(1)-Cl(2)	2.203 (6)	2.228 (7)	2.13 (4)
Cl(1)-Cl(1')	3.575 (6)	3.584 (7)	3.55 (3)
Cl(1)-Cl(2)	3.517 (6)	3.532 (7)	3.45 (3)
Cl(1)-Fe(1)-Cl(1')	111.5 (1)	111.8 (1)	111.4 (8)
Cl(2)-Fe(1)-Cl(1)	107.4 (1)	107.1 (1)	107.5 (8)

### Fe(2)/Al(2) tetrahedron

Fe(2)-Cl(3)	2.195 (4)	2.194 (4)	2.14 (1)
Fe(2)-Cl(4)	2.180 (6)	2.153 (7)	2.07 (2)
Cl(3)-Cl(3')	3.588 (7)	3.578 (7)	3.45 (1)
Cl(3)-Cl(4)	3.568 (6)	3.554 (7)	3.47 (1)
Cl(3)-Fe(2)-Cl(3)	109.7 (2)	109.3 (1)	107.9 (4)
Cl(4)-Fe(2)-Cl(3')	109.3 (1)	109.7 (1)	111.0 (4)

### Fe(3)/Mg octahedron

Fe(3)-N(1)	2.111 (9)	2.163 (3)	
Fe(3)-N(2)	2.240 (11)		
N(1)-Fe(3)-N(2)	173.5 (5)	180 (ass)	
N(1)-Fe(3)-N(1')	94.0 (4)	89.7 (1)	
N(2)-Fe(3)-N(2')	84.9 (4)		
N(1)-Fe(3)-N(2')	91.2 (4)	90.3 (1)	
N(1)-Fe(3)-N(2'')	89.6 (4)		
N(1)-N(1')	3.09 (2)	3.050 (5)	3.18 (7)
N(2)-N(2')	3.02 (2)		2.99 (6)
N(1)-N(2')	3.11 (1)	3.067 (5)	3.00 (4)
N(1)-N(2'')	3.07 (1)		3.07 (4)

For neither the Fe-Fe nor the Mg-Al compound were indications of macroscopic twinning found. An alternative description is the occurrence of polysynthetic microtwins or twinning domains as described recently by Müller (1971). The domains seem to be very small in the Mg-Al compound resulting in an OD structure (Dornberger-Schiff & Grell-Niemann, 1961). Apparently these domains are larger in the Fe-Fe compound. Nevertheless, the different e.s.d.'s of reflexions  $hkil$  and  $hh2hl$  (see Appendix) might be an indication that the structure of the Fe-Fe compound cannot be accepted as a pure case of twinning.

## III. The Ni-Ga compound

Since reflexions  $hkl$  with  $h-k=3n$  are much stronger than those with  $h-k=3n\pm 1$ , we first attempted to solve the structure of the subcell  $c'=c$ ,  $a'=b'=a/\sqrt{3}$ ,  $\alpha'=\beta'=90^\circ$ ,  $\gamma'=120^\circ$ . So far, no solution has been

found. The Patterson function indicates that its structure is closely related to that of the Fe-Fe compound.

### Discussion of the structures

Relevant interatomic distances and valence angles of the compounds Fe-Fe and Mg-Al together with the estimated standard deviations are listed in Table 5. It must be emphasized that these standard errors are grossly underestimated and do not reflect the uncertainties due to twinning or disorder. A projection of the structure on (001) is given in Fig. 2. Two independent  $\text{FeCl}_4$  tetrahedra are present at  $\frac{1}{3}$ ,  $\frac{2}{3}$ ,  $z$  and  $\frac{2}{3}$ ,  $\frac{1}{3}$ ,  $z$ . Each  $\text{FeCl}_4$  tetrahedron is repeated in the direction [001], giving rise to rows of tetrahedra. An approximate centre of symmetry relating the two tetrahedra exists on the position of Fe(II). Around this Fe(II) ion we notice an octahedron of six N-C- $\text{CH}_3$  groups.

A ligand-field spectrum of the compound Mg-Fe (type *A*) was measured by Reedijk & Groeneveld (1968*a*) and interpreted in terms of  $\text{FeCl}_4^-$  ions in tetrahedral configuration (Balt, 1967). In agreement with this conclusion two quite regular  $\text{FeCl}_4$  tetrahedra are found in the Fe-Fe structure (see Table 5).

A Mössbauer spectrum (Reedijk & Groeneveld, 1968*a*) has indicated that the Fe-Fe compound contains high-spin Fe(II) and Fe(III) ions. The small isomer shift of Fe(III) points to covalent bonds between Fe(III) and the  $\text{Cl}^-$  ions. The average Fe-Cl distance (2.18 Å) agrees nicely with the value of 2.19 Å found in the structure of  $\text{Fe}(\text{NCH})_6(\text{FeCl}_4)_2$  (Constant, Daran & Jeannin, 1970).

The average Al-Cl distance (2.13 Å) in the Mg-Al compound agrees very well with the average of Al-Cl distances (2.13 Å) in the  $\text{AlCl}_4$  tetrahedra of the compound  $\text{AlSeCl}_7$  (Stork-Blaisse & Romers, 1971).

Inspection of Table 5(*a*) shows that the octahedral coordination of Fe(II) is distorted. We notice three long and three short Fe-N distances (2.24 and 2.11 Å) and unequal N-C distances (Table 7). Far-infrared spectra were, however, interpreted by Reedijk & Groeneveld (1968*b*) in terms of equal Fe-N modes of vibration. Moreover, infrared measurements (Reedijk, Zuur & Groeneveld, 1967) could be explained on the assumption of a regular octahedral coordination of M(II). The measured magnetic susceptibility of the compound Fe-In (type *A*) (Reedijk, 1969) indicates a fairly regular octahedral configuration for the Fe(II) ion, while the small quadrupole splitting in the Mössbauer spectrum (Bancroft, Mays & Prater, 1969) of the Fe-Fe complex suggests that only small deviations from cubic symmetry are present.

We have inspected the presence of a more regular octahedron by carrying out a refinement with constraints: equal Fe(3)-N(1) and Fe(3)-N(2) distances. The atoms N(1) and N(2) kept their separate thermal parameters, but were given positions  $x$ ,  $y$ ,  $z$  and  $\bar{x}$ ,  $\bar{y}$ ,  $\bar{z}$ . After each cycle, the positional shifts of these atoms

were  $\Delta x$ ,  $\Delta y$ ,  $\Delta z$  and  $-\Delta x$ ,  $-\Delta y$ ,  $-\Delta z$  respectively. This refinement resulted in  $R_w=0.048$  and  $R=0.050$ . On account of the  $R_w$ -ratio test the model with constraints (Table 6) should be rejected.

Table 6. Numerical data for  $R_w$ -ratio tests

The symbols are explained in Hamilton (1965), significance level  $\alpha=0.005$ .

Investigated are twinned structures with  $p=0.141$ .

	I Acentric model $P\bar{3}$	II Centric model $P\bar{3}$	III Acentric model with equal Fe-N distance
$R_w$	0.0467	0.0656	0.0482
$n$	1650	1650	1650
$m$	87	45	84
$R_w(\text{II})/R_w(\text{I})=1.405$			$R_w(\text{III})/R_w(\text{I})=1.032$
$R_{42, 1563, 0.005}=1.022$			$R_3, 1563, 0.005=1.004$

In view of the uncertainties due to twinning, however, we are of the opinion that in this instance no definite choice between the two models can be made. Interatomic distances occurring in the model with constraints are likewise given in Table 5(*b*).

The Fe-N distance of the second model agrees very well with the value 2.16 Å found for the compound  $\text{Fe}(\text{NCCH}_3)_6(\text{FeCl}_4)_2$ . The longest Fe-N distance (2.24 Å) of the first model is in accord with the Fe-N length (2.25 Å) observed in the complex (HEDTA)  $\text{FeOFe}(\text{HEDTA})^{2-}$  (Lippard, Schugar & Walling, 1967).

Disregarding the carbon atoms, the molecular symmetry of the coordination about Fe(II) is  $C_3$  for the model without constraints and  $S_6$  otherwise. In view of the nearly right angles between the Fe-N bonds, the approximate symmetry of the constrained model is the cubic one  $O_h$ .

The relevant distances and angles involving the ligand atoms for models with and without equal Fe(3)-N distances are summarized in Table 7. Although the predicted standard errors for C-C and C-N bonds are about 0.015 Å, we believe that the true errors are at least 0.03 Å, since their positions near pseudo-mirror planes introduce large correlations. In both models the average values for chemically equivalent distances are C-C=1.51 Å and C-N=1.10 Å. Constant, Daran & Jeannin (1970) found the value 1.10 Å for the C-N distance in  $\text{Fe}(\text{NCH})_6(\text{FeCl}_4)_2$ . In gaseous  $\text{CH}_3\text{CN}$  (Danford & Livingston, 1955) these values are 1.47 (2) and 1.16 (3) Å, respectively. Reedijk, Zuur & Groeneveld (1967) observed a positive shift for the C-N stretching frequency in several M(II)  $(\text{NCCH}_3)_6$  ions, which is consistent with an increase of the C-N force constant (Purcell & Drago, 1966). According to Purcell (1967) the shortening of the C-N bond is in agreement with the increased force constant for the C-N stretching vibration.

We find non-linear Fe-N-C and N-C-C angles. Judged by statistical criteria, the deviation from linearity is, at least for the second type of angles, not significant, the standard deviations of these angles being about 2°.



Table 7. Distances (Å) and angles (°) involving carbon and nitrogen atoms in the Fe-Fe compound ( $p=0.141$ )

Values are given for the model without constraints (a) and the model with equal Fe(3)-N distances (b).

Columns 2 and 4 refer to average values.

	a		b	
N(1)-C(1)	1.13		1.09	
N(2)-C(2)	1.07	1.10	1.11	1.10
C(1)-C(3)	1.54		1.51	
C(2)-C(4)	1.47	1.51	1.51	1.51
Fe(3)-N(1)-C(1)	175		173	
Fe(3)-N(2)-C(2)	167	171	177	175
N(1)-C(1)-C(3)	176		176	
N(2)-C(2)-C(4)	174	175	174	175

The observed deviations may be attributed to packing effects.

It is interesting to note that non-linear C-C-N angles are found in tracyanoethylene oxide (Stucky, 1971). This author argues that both packing effects and bonding-electron distributions are responsible for the observed distortions.

The space group of  $\text{Fe}(\text{NCH})_6(\text{FeCl}_4)_2$  (Constant, Daran & Jeannin, 1970) is  $P\bar{3}$  and the lattice constants are  $a=10.29$  and  $c=6.28$  Å. A comparison of Fig. 4 in their article (used is a left-handed coordinate system) with Fig. 2 reveals that the two structures are not isomorphous.

For one thing the ligands and chlorine atoms of the hydrogen cyanide compound are not located near the planes  $\{30\bar{3}0\}$ . Moreover the relative  $z$  parameters of the atoms in the two  $\text{FeCl}_4$  tetrahedra differ considerably from ours.

All calculations were performed on the IBM 360/50 computer of the Central Computing Laboratory of the University of Leiden. We acknowledge the contributions of Mr H. P. Zoetmulder during the beginning of the investigations on the Fe-Fe compound. We appreciate stimulating discussions with Mr R. A. G. de Graaff and Dr J. Reedijk. We thank Mr A. Verhoorn (Geological Institute, University of Leiden) for taking the precession photographs. The investigations were supported with financial aid by the Netherlands Organization for the Advancement of Pure Research (ZWO).

## APPENDIX

Using equation (2) we derived the following expressions:

$$F_o(hkl) = \left\{ \frac{1-p}{1-2p} F_o'^2(hkl) - \frac{p}{1-2p} F_o'^2(khl) \right\}^{1/2} = A^{1/2} \quad (7)$$

$$\sigma[F_o(hkl)] = \left[ \left\{ \frac{1-p}{1-2p} \sigma[F_o'(hkl)]F_o'(hkl) \right\}^2 + \left\{ \frac{p}{1-2p} \sigma[F_o'(khl)]F_o'(khl) \right\}^2 \right]^{1/2} / F_o(hkl) \quad (8)$$

These equations were used for correction of all observed reflexions and the assignment of new weights. Due to errors in the measured values of  $F_o'$ , expression  $A$  can assume negative values. A boundary  $A \geq 5.30$  was used in the program. For  $A < 5.30$  the reflexion was treated as a non-observed one and assigned the value  $(|A|)^{1/2}$ . The non-observed reflexions were not changed.

Accepting trigonal symmetry and disregarding the small anomalous scattering, mean values and e.s.d.'s for a number of reflexions  $hkil$ ,  $ihkl$ ,  $kihl$ ,  $hk\bar{i}l$ ,  $ih\bar{k}l$  and  $k\bar{i}hl$  were computed. The e.s.d.'s of these reflexions were compared with the individual standard errors resulting from counting statistics, absorption and the applied polynomial expression. The e.s.d.'s are 1.65 times the standard errors for reflexions with  $h \neq k$  and 1.25 times the standard errors for reflexions with  $h = k$ .

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